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# The crystal structures of PdNi<sub>2</sub>P and Pd<sub>8</sub>Ni<sub>31</sub>P<sub>16</sub>

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#### Abstract

Two new ternary intermediate phases in the Pd–Ni–P system have been synthesized and the crystal structures have been determined by single-crystal X-ray diffraction. PdNi<sub>2</sub>P crystallizes in the MgCuAl<sub>2</sub>-type structure with the unit cell parameters  $a = 3.4708(3)$  Å,  $b = 8.4437(8)$  Å,  $c = 6.6083(5)$  Å,  $Z = 4$ , space group Cmcm. Pd<sub>8</sub>Ni<sub>31</sub>P<sub>16</sub> is tetragonal with the unit cell parameters  $a = 14.9375(4)$ Å and  $c = 5.8071(3)$ Å,  $Z = 2$ , space group  $P4<sub>2</sub>/nmc$ . The unit cell contains 110 atoms distributed on 12 crystallographic non-equivalent positions. This does not correspond to any previously reported structure.  $O$  2003 Elsevier Inc. All rights reserved.

Keywords: Transition metal phosphide; Crystal structure; X-ray diffraction

#### 1. Introduction

The ternary palladium–nickel–phosphorus system is the simplest prototype of a bulk metallic glass [\[1\].](#page-5-0) Even though the binary Pd–P and Ni–P systems contain several complex intermediate phases [\[2–9\]](#page-5-0), only a few studies of the crystalline phases in the ternary Pd–Ni–P systems have been reported. The following phases;  $Pd_{45}Ni_{34}P_{21}$  orthorhombic,  $Ni_{59}Pd_{40}P_1$  cubic F-centered,  $Ni_{14}Pd_{68}P_{18}$  orthorhombic were reported by Donovan et al. [\[10\]](#page-5-0) based on TEM investigations. The crystal structure of  $Ni_{14}Pd_{68}P_{18}$  was recently described as a solid solution of nickel in  $Pd_3P$  with a composition of  $Pd_{2.7}Ni_{0.3}P_{0.94}$  [\[11\],](#page-5-0) where the nickel atoms preferentially substitute one of the two palladium sites. A cubic high-pressure phase with composition  $Pd_2Ni_2P$  has been observed at 4 GPa pressure [\[12\]](#page-5-0). Two new ternary intermediate phases are reported in the present study.

## 2. Experimental

#### 2.1. Sample preparation

A master alloy of  $Ni<sub>2</sub>P$  was synthesized in a highfrequency furnace in an argon atmosphere, by dropping lumps of red phosphorus into a melt of nickel (nickel

rods 99.995% Johnson and Matthey) [\[13\]](#page-5-0). The sample was powdered and heat-treated in evacuated silica tubes at 800° C for 2 weeks.  $P d_x(Ni_2P)_{1-x}$  ( $x = 0.14, 0.27$ , 0:30; 0:38; 0:50; 0:53; 0:54; 0:66) were synthesized by mixing appropriate amounts of palladium (Cerac, 99.95%) and Ni<sub>2</sub>P powder. The mixtures were pressed to pellets and then heat-treated in evacuated silica tubes at  $750^{\circ}$ C for 5 days. The samples were crushed and the heat treatment was repeated at  $700^{\circ}$ C. The silica tubes were cooled in air. To improve the sharpness of the diffraction lines, the powders were stress relieved at temperatures between  $600^{\circ}$ C and  $700^{\circ}$ C, for 20–40 min. The phase stabilities at  $600^{\circ}$ C were also investigated after heat treatments for 2 weeks.

#### 2.2. X-ray powder diffraction

Phase analysis was preformed and unit cell dimensions were determined using X-ray powder diffraction techniques. Powder patterns were recorded using a Guinier–Hägg-type focusing camera with CuK $\alpha_1$  radiation and silicon as internal calibration standard  $(a = 5.431065 \text{ Å}$  at 25°C). The crystal systems were obtained using the program TREOR [\[14\]](#page-5-0).

### 2.3. Structure determination

Crystals of  $PdNi<sub>2</sub>P$  and  $Pd<sub>8</sub>Ni<sub>31</sub>P<sub>16</sub>$  were glued to glass fibers and mounted on a Bruker APEX

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<span id="page-1-0"></span>diffractometer equipped with a 2K CCD detector and using MoK $\alpha$  radiation. Data were collected using 0.2 $\degree$   $\omega$ scans and 40 s exposure time for the PdNi<sub>2</sub>P crystal;  $0.3^{\circ}$  $\omega$  scans and 60s exposure time for the Pd<sub>8</sub>Ni<sub>31</sub>P<sub>16</sub> crystal. The detector was set at  $2\theta = 28^{\circ}$  (covering  $2\theta$  = 0°–56°) for both data collections. Data reductions, cell refinements, space-group determinations and numerical absorption corrections were all done using the SAINT+program package [\[15\]](#page-5-0). Direct methods for initial starting structures and crystal structure refinements were done using the SHELXTL program package [\[15\]](#page-5-0). Due to the severe absorption in the crystals, which could not be sufficiently corrected, two new data sets were collected on a Stoe diffractometer, which enabled better absorption corrections of the intensities. New crystals of  $PdNi<sub>2</sub>P$  and  $Pd<sub>8</sub>Ni<sub>31</sub>P<sub>16</sub>$  were mounted on a Stoe & Cie four-circle diffractometer using  $M \circ K \alpha$ 

radiation. The  $PdNi<sub>2</sub>P$  data set was collected using  $\omega/2\theta$  scan; step 0.01° and step time 0.3–3 s. Data was collected up to  $2\theta = 90^\circ$ . The Pd<sub>8</sub>Ni<sub>31</sub>P<sub>16</sub> data was collected using  $\omega/2\theta$  scan; step 0.02° and step time 0.3–6 s. Data was collected up to  $2\theta = 65$ . Four standard reflections were monitored every 4 h for both crystals. The data were corrected for background [\[16\]](#page-5-0) intensities and corresponding standard deviations were corrected for time variations according to McCandlish et al. [\[17\]](#page-5-0). The complex crystal morphology was optimized using the program X-SHAPE [\[18\]](#page-5-0). The DUPALS program package was used for data reduction, numerical absorption correction and crystal structure refinements [\[19\]](#page-5-0). The function minimized was  $\sum w(F_0^2 - F_c^2)$  with  $w = 1/\sigma^2(F^2)$ . A type I isotropic extinction parameter with Lorenzian distribution of the mosaic blocks was used [\[20\]](#page-5-0). The most extinction corrected reflections were

Table 1 Experimental and refinement details for Stoe data

Experimental and relinement details for stoc data			
Empirical formula	PdNi <sub>2</sub> P	$Pd_8Ni_{31}P_{16}$	
Crystal system	Orthorhombic	Tetragonal	
Space group	Cmcm	P4 <sub>2</sub> /nmc	
Unit cell parameters determined from X-ray powder diffraction data			
a(A)	3.4708(3)	14.9375(4)	
b(A)	8.4437(8)		
c(A)	6.6083(5)	5.8071(3)	
Volume $(\AA^3)$	193.66(3)	1295.73(8)	
Z	4	$\overline{2}$	
$D_r$ (mg/m <sup>3</sup> )	8.90	8.15	
Temperature	293K	293K	
Crystal color	Metallic gray	Metallic gray	
Crystal size (mm)	$0.04 \times 0.025 \times 0.02$	$0.09 \times 0.03 \times 0.02$	
Data collection			
Diffractometer	Stoe four-circle	Stoe four-circle	
Data collection method	$\omega/2\Theta$ scan	$\omega/2\Theta$ scan	
Absorption correction	Integration	Integration	
$T_{\min}$ ( $\AA^3$ )	0.42	0.14	
$T_{\text{max}}$ $(\AA^3)$	0.60	0.51	
Linear absorption coefficient $(mm^{-1})$	28.67	28.25	
Refinement limits			
$\Theta$ (deg)	45	32.5	
h, k, l	$-6 \rightarrow h \rightarrow 6$	$-20 \rightarrow h \rightarrow 22$	
	$-13 \rightarrow k \rightarrow 12$	$-9 \rightarrow k \rightarrow 22$	
	$-15 \rightarrow l \rightarrow 15$	$-6 \rightarrow l \rightarrow 8$	
Reflections collected	2730	5648	
Independent reflections	464	1422	
Reflections in refinements $(I>4\sigma)$	378	1101	
Variable parameters	17	77	
$R_{\rm int}$	0.052	0.118	
Refinement on	$F^2$	$F^2$	
$\boldsymbol{R}$	0.034	0.068	
$R_{\rm w}$	0.042	0.050	
Extinction coefficient	6674(233)	2983(111)	
Extinction correction method	Becker and Coppens (1974), Isotropic type I,	Becker and Coppens (1974), Isotropic type I,	
	Lorentzian	Lorentzian	
Goodness-of-fit	2.04	2.83	
Largest diff. peak and hole ( $e^{-}/\text{\AA}^{3}$ )	$2.61/-1.37$	$5.48/-5.02$	

(200)  $y = 0.63$  in PdNi<sub>2</sub>P and (800)  $y = 0.77$  in  $Pd_8Ni_{31}P_{16}$ . Results and experimental details are presented in [Table 1](#page-1-0).

The low intensity of the high-angle reflections, due to the high absorption of the crystals made it difficult to determine the unit cell parameters with high accuracy. However, the unit cell parameters obtained from the single crystals are within experimental errors in agreement with the X-ray powder diffraction results. All unit cell dimensions reported are based on X-ray powder diffraction data.

#### 3. Results and discussion

The crystal structures of two new ternary intermediate phases  $PdNi<sub>2</sub>P$  and  $Pd<sub>8</sub>Ni<sub>31</sub>P<sub>16</sub>$  have been determined. PdNi<sub>2</sub>P was found in samples  $Pd_x(Ni_2P)_{1-x}$  (x =  $0.38-0.66$ ). As judged from X-ray powder diffraction patterns, small variations in unit cell volumes, less than  $1.5 \text{ Å}^3$ , indicate a small range of homogeneity. Samples,  $Pd_x(Ni_2P)_{1-x}$   $(x = 0.14-0.30)$ , prepared at 700°C, showed diffraction lines from a common phase. The unit cell parameters from different samples were all the same within experimental errors. This phase is stable at 700°C but disproportionates at 650°C to  $Ni<sub>2</sub>P$ ,  $Ni<sub>12</sub>P<sub>5</sub>$ and an unidentified phase. The X-ray powder diffraction patterns of several samples contained diffuse extra lines, which did not improve in quality despite several attempts to stress relieve the powders. The phase relations of the ternary system were not determined due to the presence of additional phases.

## 3.1. PdNi<sub>2</sub>P

PdNi<sub>2</sub>P has a small range of homogeneity, with palladium substitutions on the nickel position. The unit cell parameters for  $Pd(Ni_{1-x}$   $Pd_x)_2P$  ( $x = 0.06$ ) were determined from X-ray powder diffraction patterns to  $a = 3.4708(3)$  Å,  $b = 8.4437(8)$  Å,  $c = 6.6083(5)$  Å with a unit cell volume of 193.66(3)  $A^3$  and  $Z = 4$ , space group Cmcm, [Table 1.](#page-1-0)  $PdNi<sub>2</sub>P$  crystallizes in the  $MgCuAl<sub>2</sub>$ -type structure [\[21\].](#page-6-0) This structure type has three crystallographic sites, with aluminum at the 8f-position, magnesium at  $4c$  and copper at  $4c$ . In PdNi2P, the nickel atoms occupy the 8-fold aluminum site and the palladium and phosphorus atoms occupy the 4c magnesium and copper positions, respectively. The refinements indicated a small substitution of palladium on the nickel site and full palladium occupancy on the other metal position. The structural parameters calculated in the last single-crystal refinement cycle are shown in Table 2. The crystal structure is shown in projection on the bc plane in Fig. 1. The MgCuAl2-type structure is a ternary ordered derivative of the  $Re<sub>3</sub>B$ -type structure. The atomic arrangement is Table 2

Fractional atomic coordinates and equivalent isotropic displacement parameters for the PdNi<sub>2</sub>P structure

Atom	Position $x \, y$			Occ $(\% )$	$U_{\text{eq}}(\AA^2)$
$Ni/Pd1$ 8f Pd <sub>2</sub>	4c	$0$ 0.14510(5) 0 $0 \quad 0.44552(4)$	1/4	$94.8(5)/5.2(5)$ 0.0075(1) - 100	0.00801(8)
P1	4c	$0 \quad 0.7321(2)$	$1/4$ 100		0.0071(2)



Fig. 1. The crystal structure of PdNi2P.

easily described in terms of columns of filled trigonal prisms along the a-axis, with one phosphorus atom in the middle of each prism. Each phosphorus atom coordinates nine metal atoms. Four nickel and two palladium atoms are situated at the vertices of the prism, two nickel and one palladium atoms are found outside the rectangular faces.

The Ni–P distances are shorter than the Pd–P distances. Interatomic distances shorter than  $3.0 \text{\AA}$  are listed in [Table 3](#page-3-0). Iso-structural compounds have been

<span id="page-3-0"></span>reported in ternary aluminum, gallium, tin and indium compounds [\[22–25\]](#page-6-0). The crystal structure of  $Pd_6P$  [\[3\]](#page-5-0) is a superstructure also related to the  $Re<sub>3</sub>B$ -type structure. Every second trigonal prism within the columns accommodates one phosphorus atom and the columns are stacked in a certain order to each other.

## 3.2.  $Pd_8Ni_{31}P_{16}$

 $Pd_8Ni_{31}P_{16}$  has a tetragonal unit cell with the unit cell parameters  $a = 14.9375(4)$  Å and  $c = 5.8071(3)$  Å, determined from X-ray powder diffraction pattern, space group  $P4_2/mnc$ , [Table 1](#page-1-0). The unit cell of tetragonal  $Pd_8Ni_{31}P_{16}$  contains 110 atoms distributed on 12 crystallographic non-equivalent positions. The crystal structural parameters, from the last crystal structure refinement cycle, are listed in Table 4. [Table 5](#page-4-0) shows interatomic distances shorter than  $3.0 \text{ Å}$ . The metal atoms form capped trigonal prisms, which accommodate the phosphorus atoms P2 and P3, respectively. The P2 and P3 thus have nine close metal neighbors, six at

Table 3 Interatomic distances in PdNi<sub>2</sub>P  $(\AA)$ 

Ni/Pd1	P <sub>1</sub>	2.2606(8)
	2 P1	2.2868(5)
	Ni/Pd1	2.5906(7)
	2Pd2	2.7440(4)
	2Pd2	2.7627(4)
	Pd <sub>2</sub>	2.8482(5)
Pd <sub>2</sub>	P <sub>1</sub>	2.419(1)
	2P1	2.501(1)
	2Ni/Pd1	2.7440(4)
	4Ni/Pd1	2.7627(4)
	Ni/Pd1	2.8482(5)
P <sub>1</sub>	2Ni/Pd1	2.2606(8)
	4Ni/Pd1	2.2868(5)
	Pd2	2.419(1)
	2Pd2	2.501(1)

Table 4

Fractional atomic coordinates and equivalent isotropic displacement parameters for the Pd<sub>8</sub>N<sub>131</sub>P<sub>16</sub> structure

the vertices of the prism and three situated outside each of the quadrilateral faces. The prisms are linked together to form columns of prisms pairs along the c-axis (see [Fig. 2 and 3\)](#page-5-0). The 10 metal neighbors surrounding the third phosphorus atom, P1, cannot be described in terms of a regular polyhedron. The P1 coordination polyhedra share one edge and buckled rings are formed around the columns of the capped P2 and P3 trigonal prisms. The pure nickel sites in  $Pd_8Ni_{31}P_{16}$  have four phosphorus neighbors except for Ni7 that coordinates three phosphorus atoms.

Two metal sites have mixed occupancies of palladium and nickel, Ni/Pd5 and Pd/Ni1. The Ni/Pd5 site, which is substituted by palladium to almost 50%, has four phosphorus close neighbors. Pd/Ni1, which is occupied to 80% by palladium, has five phosphorus neighbors. The shortest metal–phosphorus distance is  $2.17 \text{ Å}$ between Ni7 and P1. Such a short Ni–P distance has previously been reported for the rhombohedral  $\alpha$ -Ni<sub>8</sub>P<sub>3</sub> [\[9\]](#page-5-0). The shortest Pd–P distances are 2.302, 2.327 and 2.341 A for  $Ni/Pd5$ ,  $Pd/Ni1$  and  $Pd2$ , respectively. There are several metal-rich palladium phosphides but  $Pd_7P_3$ and  $Pd_3P_{0.8}$  have the metal-to-phosphorus ratio closest to  $Pd_8Ni_{31}P_{16}$ . The shortest interatomic distances between phosphorus and palladium in these structures are 2.18 and 2.22 Å. Palladium has a larger metal radius than nickel so the shortest Me–P distance increases with increased occupancy of palladium at a particular position.

#### 3.3. Structural properties in the Pd–Ni–P system

 $Pd_3P$ , crystallizing in the cementite structure,  $Fe_3C$ type, has previously been reported to form a solid solution with nickel. The cementite structure has two crystallographic metal positions: one surrounded by three and one by two phosphorus atoms. Nickel substitutes on the palladium position coordinating three phosphorus atoms with the longest metal–phosphorus distances.



Table 5 (continued)

<span id="page-4-0"></span>Table 5 Interatomic distances in  $Pd_8Ni_{31}P_{16}$  (Å)

Ni1	4P3	2.246(3)
	4Ni4	2.544(1)
	4Pd/Ni1	2.7019(9)
Ni2	P <sub>2</sub>	2.206(3)
	P <sub>3</sub>	2.224(3)
	2P1	2.445(2)
	Ni7	2.469(2)
	2Ni4	2.498(2)
	2Ni/Pd5	2.556(2)
	2Ni6	2.666(2)
	Pd/Ni1	2.804(2)
Ni3	2P1	2.241(2)
	2P1	2.459(3)
	Ni4	2.487(2)
	2Ni6	2.508(2)
	2Ni/Pd5	2.784(2)
	2Ni6	2.856(2)
	2Ni3	2.9460(3)
Ni4	2P3	2.265(2)
	2P1	2.345(2)
	Ni3	2.487(2)
	2Ni2	2.498(2)
	2Ni6	2.528(1)
		2.544(1)
	Ni1	
	2Pd/Ni1	2.7706(8)
Ni/Pd5	P1	2.302(3)
	P <sub>2</sub>	2.367(2)
	P <sub>2</sub>	2.409(2)
	Ni2	2.556(2)
	Ni7	2.562(1)
	Ni7	2.707(2)
	Ni6	2.732(2)
	Ni3	2.784(2)
	P1	2.841(2)
	Pd <sub>2</sub>	2.842(1)
	2Ni/Pd5 Pd <sub>2</sub>	2.949(2) 2.989(1)
Ni6	P1	2.249(2)
	P3	2.297(2)
	P1	2.410(3)
	Ni3	2.508(2)
	Ni4	2.528(1)
	Ni7	2.564(2)
	P <sub>2</sub>	2.630(2)
	Ni2	2.666(2)
	Ni/Pd5	2.732(2)
	Ni6	2.753(2)
	Pd/Ni1	2.757(2)
	Ni3	2.856(2)
Ni7	P <sub>2</sub>	2.174(3)
	2P1	2.249(3)
	Ni2	2.469(2)
	2Ni/Pd5	2.562(1)
	2Ni6	2.564(2)
	2Ni/Pd5	2.707(2)
	Pd/Ni1	2.833(2)
	Pd <sub>2</sub>	2.965(2)
Pd/Ni1	P <sub>3</sub>	2.328(3)

2P3 2.559(2)



In both structures, at least one site can be occupied alternately by nickel or palladium.

In the orthorhombic structure, nickel substitutes on one palladium site and in the tetragonal structure two metal positions have a mixed occupancy of palladium and nickel, Ni/Pd5 and Pd/Ni1. The nickel atoms prefer the metal positions with high coordination of phosphorus. The Pd1/Ni1 position has five phosphorus neighbors in the tetragonal structure and in the cementite structure nickel prefers the site with three phosphorus neighbors. The metal-to-phosphorus distances increase at higher palladium content in accordance with the larger radius of palladium.

## 4. Conclusion

The crystal structures of two new ternary intermediate phases have been determined in the Pd–Ni–P system.  $PdNi<sub>2</sub>P$  crystallizing in the MgCuAl<sub>2</sub>-type structure has

<span id="page-5-0"></span>

Fig. 2. Capped trigonal prisms occupied by (a) P2 light grey and P3 dark gray, and (b) P1 in the crystal structure of  $Pd_8Ni_{31}P_{16}$ .



Fig. 3. Columns of capped trigonal prisms occupied by (a) P2, and (b)P3.

a small range of homogeneity with some substitution of palladium on the nickel site. A high-temperature phase with the approximate composition  $Pd_8Ni_{31}P_8$  is stable above  $700^{\circ}$ C. The range of homogeneity is very narrow, since no variations of the unit cell were detected.

All sites mainly occupied by nickel are closer to the phosphorus atoms than the palladium positions. Nickel occupies the metal sites with the highest coordination of phosphorus. Higher palladium occupancies on a metal position increase the metal–phosphorus distance due to its larger radius compared to nickel.

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